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# Divergent growth of poly(amidoamine) dendrimer-1 like branched polymers at the reducing end of 2 cellulose nanocrystals 3 4

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#### 6 ABSTRACT

5

7 This paper presents the growth of dendritic polymers at the reducing ends of cellulose 8 nanocrystals by the "grafting from" approach. We took advantage of the chemically 9 differentiated ends of cellulose nanocrystals to specifically synthesize dendrimers at their 10 reducing end by the divergent approach. We used acid-amine coupling reactions in aqueous 11 media to synthesize the carboxylic acid- or amine-terminated poly(amidoamine) dendrimers. The 12 growth of dendrimer generations was monitored by UV and FTIR spectroscopies, and we 13 successfully introduced up to 4 generations. The dendrimer growth at reducing ends was 14 demonstrated by the nanocrystal adsorption driven by the peripheral amino groups onto gold 15 surfaces. Hence, the results from quartz crystal microbalance with dissipation (QCM-D) pointed 16 to a rather upright orientation of the dendrimer-modified cellulose nanocrystals. As the 17 generation increased, the adsorbed layers appeared to be more flexible, which demonstrated that 18 the functionality at the reducing end can successfully tune the properties of cellulose 19 nanocrystals.

20 **Keywords**: tunicate cellulose nanocrystals; reducing end functionalization; poly(amidoamine) 21 dendrimer; oriented adsorption.

1

#### 23 **1. Introduction**

24 Cellulose nanocrystals (CNC) have gained prominence as bio-based materials for a wide range 25 of applications (Charreau, H., Cavallo, E., & Foresti, M. L., 2020) thanks to their mechanical and 26 barrier properties, liquid crystal behavior, optical properties, and amphiphilic characteristics 27 (Eichhorn, S. J., 2011; Heise, K. et al., 2020; Lin, N., Huang, J., Chang, P. R., Anderson, D. P., 28 & Yu, J., 2011). CNC are obtained from cellulose fibers by acid hydrolysis that preferentially 29 solubilizes the disordered regions of cellulose, resulting in rod-like crystallites with cross-30 sections in the nanoscale (5-20 nm) and aspect ratio values in the 5-50 range (Sacui, I. A. et al., 31 2014). One of the particularities of cellulose is the orientation of the chains within the crystal 32 structure yielding two chemically differentiated ends. Indeed, the parallel arrangement of 33 cellulose chains in the cellulose I allomorph results in C1 reducing ends (aldehyde 34 functionalities) at one end of the nanocrystal, and C4 hydroxyl groups at the opposite end. 35 Taking advantage of this chemical polarity (hydroxyl versus aldehyde), the selective 36 modification of CNC reducing ends has recently attracted much attention (Heise, Katja et al., 37 2020; Tao, H., Lavoine, N., Jiang, F., Tang, J., & Lin, N., 2020) However, there are still very 38 few studies focusing on the growth of functionalities at reducing ends by the "grafting from" 39 approach (Delepierre, G. et al., 2021; Risteen, B. et al., 2018; Zoppe, J. O. et al., 2017). 40 Generally, the polymerization initiator is introduced at the reducing end and the subsequent 41 addition of the monomers yields to the polymeric functionality. This article describes for the first 42 time the synthesis of poly(amidoamine) dendrimers at the CNC reducing ends by the "grafting 43 from" approach. Our strategy involves straightforward acid-amine coupling reactions, and CNC 44 act as solid support to immobilize the core and grow dendrimer generations.

45 Dendrimers are regular hyperbranched polymers synthesized from a multi-functional core by 46 repeated cycles of synthesis. They are characterized by the number of branches (generation) and 47 the peripheral functional groups (amino, carboxyl, hydroxyl or hydrocarbon groups). Therefore, 48 dendrimers display multiple peripheral functional groups giving rise to multivalencies and a 49 tunable size depending on dendritic generation. Hence, dendrimers have found many 50 applications in different fields as nanoscaffolds for targeted drug delivery, and templates for the 51 fabrication of metal nanoparticles (Guo, Y. Q., Shen, M. W., & Shi, X. Y.; Nikzamir, M., 52 Hanifehpour, Y., Akbarzadeh, A., & Panahi, Y.). The grafting of dendrimers on cellulose offers 53 many advantages resulting from its promising properties such as biodegradability, 54 biocompatibility, and availability. Poly(propylene imine) (PPI) and poly(amidoamine) 55 (PAMAM) dendrimers have been grown on the CNC surface by iterative Michael additions 56 (Golshan, M., Salami-Kalajahi, M., Roghani-Mamaqani, H., & Mohammadi, M., 2017; Hassan, 57 M. L., 2006; Tehrani, A. D. & Basiryan, A., 2015; Wang, Y. C. & Lu, Q. Y., 2020). In those 58 studies, dendrimers covered the whole CNC surface and changed significantly their surface 59 properties, inducing CNC aggregation. In this work, we specifically synthesized dendrimers at 60 the CNC reducing end, thus retaining the CNC surface properties while adding the dendritic 61 functionalities at one end. As previously showed, the introduction of functionalities at the CNC 62 reducing ends allows the supramolecular assembly by this end, and tunes the responsiveness 63 (Chemin, M., Moreau, C., Cathala, B., & Villares, A., 2020; Chemin, Maud, Moreau, Céline, 64 Cathala, Bernard, & Villares, Ana, 2020; Lokanathan, A. R. et al., 2013; Villares, A., Moreau, 65 C., & Cathala, B., 2018). This work opens new perspectives for CNC applications, by their 66 patchy modification to achieve the desired functionality at the reducing end.

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- 70
- 71 **2. Materials and methods**
- 72 2.1. Cellulose nanocrystals preparation

Cellulose nanocrystals (t-CNC) were obtained from tunicates (*Styela clava*) collected on the coast of Quimiac (France) and prepared and characterized as previously described (Chemin, M. et al., 2020; Chemin, Maud et al., 2020; Villares, A. et al., 2018). The surface charge density, calculated by conductometric titration (TIM900 titration manager and CDM230 conductimeter), was  $0.022\pm 0.001$  mmol g<sup>-1</sup>.

78 2.2. Materials

*N*-(3-dimethylaminopropyl)-*N*-ethylcarbodiimide hydrochloride (EDC), *N*-hydroxysuccinimide
(NHS), hexamethylenediamine (HMDA) (98%), trimesic acid (TMA) (95%), sodium chlorite
(NaClO<sub>2</sub>), acetic acid, potassium chloride (KCl) and potassium hydroxide (KOH) were
purchased from Sigma-Aldrich (France) and were used without further purification. Dialysis
membrane, molar mass cut off 12-14 kDa, was purchased from Spectrum Laboratories Inc.
Water was purified by Millipore Milli-Q purification system (18.2 MΩ).

- 85 *2.3. Methods*
- 86 2.3.1. Introduction of amine moieties.

Prior to the first step of functionalization, the aldehyde groups on the reducing end of t-CNCs were oxidized to carboxylic acid groups as previously described (Chemin, M. et al., 2020; Chemin, Maud et al., 2020; Villares, A. et al., 2018). Briefly, t-CNC (200 mL at 6.9 g  $L^{-1}$ ) were oxidized by the addition of 33.7 mmol of NaClO<sub>2</sub>, and the pH was adjusted to 3.5 using 1 M acetic acid, followed by stirring for 20 h at room temperature. The reaction mixture was purified
by centrifugation (20000 g, 20 °C, 60 min) followed by dialysis (molar mass cut off 12-14 kDa)
against milli-Q water for 10 days. Dispersions were stirred in water for 24 h to avoid nanocrystal
aggregation.

95 Afterwards, amine moieties were introduced using the NHS/EDC amide coupling with 96 hexamethylenediamine (Lokanathan, A. R., Lundahl, M., Rojas, O. J., & Laine, J., 2014). A 2 g  $L^{-1}$  suspension of oxidized t-CNC was degassed by bubbling nitrogen for 20 min, and NHS (50 97 98 mmol) was added, followed by EDC (5 mmol), and the pH was checked to be close to 6.5. 99 Appropriate amounts of KCl were added such that the final suspension was 0.5 M. 100 Hexamethylenediamine was added in large excess (5 mmol) to avoid the grafting of two t-CNCs 101 on the same molecule. Then, pH was checked to be above 9.2. The reaction was incubated under 102 magnetic stirring at room temperature for 24 h. Amine-functionalized t-CNCs were purified by 103 centrifugation (20000 g, 60 min, 20 °C) and then dialyzed (molar mass cut off 12-14 kDa) 104 against Milli-Q water for 5 days. After dialysis, dispersions were stirred in water for 24 h to 105 avoid nanocrystal aggregation.

### 106 2.3.2. Introduction of acid moieties

107 A similar procedure was applied to graft trimesic acid on the amine-functionalized t-CNCs. 108 Appropriate amounts of KCl were added to the suspension of amine-functionalized t-CNCs such 109 that the final suspension was 0.5 M and the dispersion was stirred under nitrogen bubbling for 20 110 min. A 0.95 g L<sup>-1</sup> solution of trimesic acid in water was degassed by bubbling nitrogen for 5 min, 111 NHS were added (3 mmol), followed by EDC (15 mmol) under stirring and bubbling nitrogen. 112 Both suspensions were then mixed with a final t-CNC concentration of 1.5 g L<sup>-1</sup> (5 mmol of 113 amine groups). pH was checked to be above 9.2, and the reaction was incubated under magnetic 114 stirring at room temperature for 24 h. Acid-functionalized t-CNCs were purified by 115 centrifugation (20000 g, 60 min, 20 °C) and then dialyzed (molar mass cut off 12-14 kDa) 116 against Milli-Q water for 5 days. Dispersions were stirred in water for 24 h to avoid nanocrystal 117 aggregation. After grafting, hybrid nanoparticles are denoted t-CNC-G*x*, *x* being the generation. 118 When *x* is a decimal number, the peripheral groups are the carboxylic acid, and when *x* is an 119 integer, the t-CNC peripheral groups are the amine functionalities. The step-by-step synthesis 120 was repeated consecutively up to the synthesis of the fourth generation t-CNC-G4.

121 2.4. Characterization

122 2.4.1. Scanning Transmission Electron Microscopy (STEM)

123 t-CNC dispersions were diluted to  $0.2 \text{ g L}^{-1}$  in KOH 1 mM and deposited on freshly glow-124 discharged carbon-coated electron microscope grids (200 mesh, Delta Microscopies, France). 125 Water excess was removed by blotting (Whatman filter paper). The grids were dried overnight in 126 air at ambient temperature and then coated with platinum layer by an ion-sputter coater 127 (thickness = 0.5 nm). The grids were observed with a Quattro Scanning electron microscopy 128 (Thermo Scientific) with a STEM detector, working at 10kV.

129 2.4.2. Elemental analysis. t-CNCs were freeze-dried and the resulting powder (10 mg) was 130 placed in tin capsules for analysis. The total C, N, and S contents of t-CNCs were determined by 131 a C/N/S analyzer (Vario Micro cube, Elementar) coupled to a gas chromatographic system and a 132 thermal conductivity detector (TCD). Nitrogen content was determined by the Dumas' method 133 combustion (Shea, F. & Watts, C. E., 1939).

134 2.4.3. Infrared spectroscopy

135 Infrared spectra were obtained from KBr pellets containing freeze-dried t-CNC and t-CNC-Gx

136 samples placed directly in a Nicolet iS50 FTIR spectrometer (Thermo Scientific) in absorbance

mode. All spectra were recorded using OMNIC software with a 4 cm<sup>-1</sup> resolution after 200
continuous scans from 400 to 4000 cm<sup>-1</sup>. All samples were acidified with HCl 100 mM in order
to protonate carboxylic acids.

140 2.4.4. UV-visible spectroscopy

141 UV-visible spectra were obtained from t-CNC-Gx suspensions at 0.3 g L<sup>-1</sup> in KOH 1mM using a 142 quartz cell placed directly in a SPECORD S 600 spectrometer (Analytik Jena GmbH). All 143 spectra were recorded using WinASPECT software set to absorbance mode after 10 144 accumulations and scanning from 200 to 1000 nm. A reference sample consisting of t-CNC at 145 0.3 g L<sup>-1</sup> in KOH 1mM mixed with trimesic acid at 0.001 g L<sup>-1</sup> in KOH 1 mM was used for 146 comparison.

147 2.4.5. Thermal degradation

148 Thermogravimetric analysis (TGA) were performed under a nitrogen atmosphere using a TA 149 instruments TGA2050. Samples (5 mg) were equilibrated at 60°C for 10 minutes and heated 150 from 60 to 600 °C at a heating rate of 5 °C min<sup>-1</sup>.

151 2.4.6. Surface Plasmon Resonance (SPR)

The SPR measurements were performed with a Biacore x100 (GE Healthcare, Sweden) with a continuous flow system. The SPR technique is based on the phenomenon of total internal reflection of light. The adsorption of dendrimer-modified t-CNC onto the sensor surface produces a shift in the optical resonance properties of the sensor, which results in an increase of the SPR angle where the reflected light intensity is in its minimum (Schoch, R. L. & Lim, R. Y. H., 2013). All measurements were carried out using gold sensors at 20 °C with a 0.3 g L<sup>-1</sup> dispersion of t-CNC in KOH 1mM at 10  $\mu$ L min<sup>-1</sup>. The amount of adsorbed dendrimer-modified 159 t-CNC,  $\Gamma$ , was then calculated from the SPR angle shift,  $\Delta\theta$ , according to the following 160 expression (Kontturi, K. S., Kontturi, E., & Laine, J., 2013):

161 
$$\Gamma = \frac{l_d}{2} \frac{\Delta \theta}{m(dn/dc)}$$
(1)

Where  $l_d$  is the characteristic evanescent electromagnetic field decay length (281.2 nm), *m* is the sensitivity factor for the sensor (102.5° per RIU) obtained after calibration, and *dn/dc* is the refractive index increment (0.15 mL g<sup>-1</sup>). Results are expressed as the mean of two experiments.

165 2.4.7. Quartz crystal microbalance with dissipation monitoring (QCM-D)

166 QCM-D measurements were performed with a Q-Sense E4 instrument (AB, Sweden) using a 167 piezoelectric AT-cut quartz crystal coated with gold electrodes on each side (QSX301, Q-Sense), as previously described (Chemin, Maud et al., 2020; Villares, A. et al., 2018). All measurements 168 were carried out using the QCM flow cell modules at 20 °C with a 0.3 g L<sup>-1</sup> dispersions of t-169 CNC in KOH 1 mM at 100  $\mu$ L min<sup>-1</sup>. Frequency ( $\Delta f_n/n$ ) and ( $\Delta D_n$ ) dissipation changes were 170 171 simultaneously registered at 5 MHz fundamental resonance frequency and its several overtones 172 as a function of time. Any material adsorbed on the crystal surface induces a decrease of the 173 resonance frequency. If the adsorbed mass is evenly distributed, rigidly attached and small 174 compared to the mass of the crystal,  $\Delta f$  is directly proportional to the adsorbed mass ( $\Delta \Gamma$ ) using 175 the Sauerbrev's equation (Sauerbrev, G., 1959):

176 
$$\Delta \Gamma = -C\frac{\Delta f}{n} \tag{2}$$

where *C* is the constant for the mass sensitivity of the quartz crystal (0.177 mg m<sup>-2</sup> Hz<sup>-1</sup> at  $f_0 = 5$ MHz) and *n* is the overtone number. 179 Dissipation signals were recorded simultaneously to frequency shifts and they provide a measure

180 of the frictional losses due to the viscoelastic properties of the adsorbed layer. The overtone n = 5

181 was used for comparing the adsorption of unmodified and modified t-CNCs.

182

#### 183 **3. Results**

#### 184 *3.1. Dendritic synthesis at t-CNC reducing end*

185 Cellulose nanocrystals can be viewed as rigid rod-like crystals with two chemically differentiated 186 ends. Hence, reducing ends display aldehyde groups whereas the non-reducing ends present the 187 less reactive hydroxyl groups. This chemical asymmetry allows the selective introduction of 188 functionalities at the reducing end of cellulose nanocrystals (Lin, F., Cousin, F., Putaux, J.-L., & 189 Jean, B., 2019; Lokanathan, A. R. et al., 2013; Villares, A. et al., 2018). In this work, we have 190 used tunicate cellulose nanocrystals (t-CNC) as a solid support for the divergent synthesis of 191 poly(amidoamine) dendrimers at the t-CNC reducing ends. Tunicate cellulose nanocrystals were 192 selected because they are quasi-perfect monocrystals and, besides their narrow distribution in 193 cross-section and length dimensions, they have a rectangular 8.8 nm  $\times$  18.2 nm cross-sectional 194 shape (Terech, P., Chazeau, L., & Cavaille, J. Y., 1999). As the monomers, we selected a 195 diamine, the hexamethylenediamine (HMDA), and a tricarboxylic acid, the trimesic acid (TMA). 196 Therefore, dendrimer growth was driven by TMA, whereas the HMDA grafting only changed 197 the peripheral functional groups.

198 Our synthetic approach first involved the oxidation of t-CNC aldehyde groups into carboxylic 199 acids and the subsequent dendrimer growth by the N-(3-dimethylaminopropyl)-N-200 ethylcarbodiimide hydrochloride (EDC) chemistry. Fig. **1** shows the theoretical dendrimer 201 generation growth at the reducing end of cellulose nanocrystals. EDC-activated carboxylic acids

202 were transformed into amine functionalities by their reaction with hexamethylenediamine. 203 HMDA was added in high excess to avoid cross-linking. Furthermore, reactions were carried out 204 above pH 9.2 and at high ionic strength (KCl 0.5 M) to screen t-CNC interactions and avoid 205 unspecific adsorption of chemicals. The second step was the reaction of amine-functionalized t-206 CNC with trimesic acid. Reaction was carried out under similar conditions, TMA was EDC-207 activated and added to amino-t-CNC dispersed in high ionic strength solutions (KCl 0.5 M) at 208 pH 9.2. This two-step procedure was repeated 4 times to achieve the amine-terminated generation 4 (t-CNC- G4). 209



210

- Fig. 1. Scheme of the divergent dendrimer synthesis on the reducing ends of t-CNC by the
- successive grafting of HMDA and TMA by EDC/NHS activation. In the scheme, t-CNC are
- 213

represented as green rods. Dimensions of t-CNC, HMDA and TMA are not scaled.

214

215 Generally, poly(amidoamine) dendrimers are prepared by serial repetitions of the Michael 216 addition of amine-functionalized CNCs (Golshan, M. et al., 2017; Hassan, M. L., 2006; Tehrani, 217 A. D. & Basiryan, A., 2015). Recently, Wang et al. have described a protocol including a 218 Michael addition followed by amination in methanol (Wang, Y. C. & Lu, O. Y., 2020). 219 However, in all cases, reactions were performed in methanol/water mixtures. Our approach 220 involved successive amidation reactions with EDC/NHS activation and did not require the use of 221 organic solvents or high temperatures. Reactions were entirely performed in aqueous media at 222 room temperature. Purification and removal of soluble chemicals after each generation growth 223 was facilitated by t-CNC precipitation. The high ionic strength (0.5 M) favored nanocrystal 224 precipitation and then, salt was removed by dialysis. The recovered t-CNC masses after each 225 reaction were higher than 99% for both the first oxidation and the successive EDC/NHS 226 coupling. t-CNC-Gx were stirred overnight in KOH 1 mM to avoid nanocrystal aggregation 227 arising from the interaction between positively charged amino groups and the negative sulfate 228 groups in the surface of t-CNC. As Fig. 2 shows, the morphology of t-CNC was not altered by 229 the dendritic growth at their reducing ends, the t-CNC exhibited the typical and expected rod-like 230 shape. The STEM images did not show significant differences between when the dendrimer was 231 amino- or acid-terminated, and no significant association between nanocrystals was observed due 232 to dendrimer growth. Hence, t-CNC can be viewed as a solid support to hold the dendrimer so 233 the excess of chemicals can be easily removed by centrifugation and dialysis.





Fig. 2. STEM images of native t-CNC, amine-terminated (t-CNC-G0), the first acid- and amineterminated (G0.5 and G1, respectively) and last acid- and amine-terminated (G3.5 and G4, respectively) dendrimer generations grown on t-CNC reducing ends.

239 The elemental compositions of t-CNC-G4 compared to native t-CNC (Table 1) revealed a 240 significant higher nitrogen content for t-CNC-G4. According to the carbon-nitrogen (C/N) ratio, 241 for t-CNC-G4 there was a nitrogen atom for each 45 glucose units. As the dendrimer G4 contains 242 62 nitrogen atoms, there was one dendrimer per 2800 glucose units. The degree of 243 polymerization (DP) of cotton cellulose nanocrystals is between 150 and 325 (Potthast, A. et al., 244 2015); therefore, considering an elemental fibril formed by 16 cellulose chains and a high DP for 245 tunicate nanocrystals, there was approximately 1.6 dendrimers for each cellulose nanocrystal. 246 This high degree of substitution could be explained by the introduction of several dendrimer

functionalities at the reducing end, or by some degree of chain breakdown during reactions thatcreated new reducing ends.

249

250 Table 1

Elemental compositions and elemental C/N ratios for native t-CNC and t-CNC-G4.
Determinations were made in triplicate and results are expressed as the mean ± standard deviation.

	N (%)	C (%)	S (%)	C/N
t-CNC	$0.002 \pm 0.002$	$43.925 \pm 0.686$	$0.485 \pm 0.169$	$9050 \pm 764$
t-CNC-G4	$0.186 \pm 0.002$	$43.090\pm0.127$	$0.305 \pm 0.064$	$232 \pm 1$

254

Fig. 3 shows the FTIR spectra of native t-CNC and t-CNC-Gx for the first generations (G0.5 and
G1) and the last ones (G3.5 and G4). The FTIR spectrum of native t-CNCs (green bottom trace)
showed the characteristic bands of cellulose nanocrystals (Zhao, Y. & Li, J., 2014; Zhao, Y. D.,
Zhang, Y. J., Lindstrom, M. E., & Li, J. B., 2015). Upon dendrimer growth, the FTIR profile was
not modified, which indicated that the crystalline order of t-CNC was not altered by reducing end
modification.



261

Fig. 3. FTIR spectra in absorbance of native t-CNC and the first acid- and amine-terminated
 (G0.5 and G1, respectively) and last acid- and amine-terminated (G3.5 and G4, respectively)
 dendrimer generations grown on t-CNC reducing ends.

266 The dendrimer growth was suggested by the increase in the C=O stretching band. For acidterminated generations (G0.5 and G3.5), FTIR spectra showed a new peak at 1720 cm<sup>-1</sup>, which 267 268 corresponded to the C=O stretching band of COOH groups. The intensity of this band increased 269 with generation, which pointed at dendritic growth. In the case of amine-terminated generations 270 (G1 and G4), the peak from COOH had decreased in intensity compared to their precursors, 271 indicating their partial transformation into amide forms. The spectra showed an increase in the shoulder intensity at 1700 cm<sup>-1</sup>, which can be ascribed to the C=O stretching from amide I 272 bonding at 1645 cm<sup>-1</sup> (Benkaddour, A., Journoux-Lapp, C., Jradi, K., Robert, S., & Daneault, C., 273 274 2014). However, this small signal from the amide bonds would be overlapped with the band at 1635 cm<sup>-1</sup> corresponding to the symmetric deformation vibration of adsorbed water molecules 275 276 (Schwanninger, M., Rodrigues, J. C., Pereira, H., & Hinterstoisser, B., 2004). On the other hand,

the bands of the C-N stretching and N-H deformation from amide II at 1545 cm<sup>-1</sup> were not
detected.

Dendritic growth was also demonstrated by the increase in the UV absorption band at 228 nm corresponding to TMA. Fig. **4** shows the absorbance spectra of a mixture of native t-CNC and TMA, and t-CNC-G0, t-CNC-G2 and t-CNC-G4.



282

**Fig. 4**. Absorbance spectra of free TMA in a native t-CNC dispersion, and the dendrimer

generations t-CNC-G0, t-CNC-G2 and t-CNC-G4.

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The grafting of TMA on t-CNC resulted in an increasing baseline at low wavelengths possibly due to slight aggregation of nanocrystals, as well as the appearance of a shoulder at 280 nm ascribed to the TMA absorption band. When the generations increased, the shoulder at 280 nm become progressively more pronounced and, for the last generation, t-CNC-G4, a well-resolved band was observed. The possibility of unspecific adsorption of chemicals (HMDA and TMA) on the cellulose surface was discarded because in all cases, HMDA and TMA were added in high excess and t-CNC were thoroughly washed for their removal after reactions. Therefore, if chemicals were adsorbed on t-CNC we would not see the step-by-step increase of FTIR and UVabsorption bands.

295 t-CNC-G4 was further studied by thermogravimetric analysis to compare with native t-CNC 296 (Fig. 5). The thermal analysis showed a two-step thermal degradation process, as previously 297 described for cellulose nanocrystals (Lin, N. & Dufresne, A., 2014). First, the pyrolysis is 298 catalyzed by the sulfate groups present on the t-CNC surface and then residues are carbonized 299 (Wang, N., Ding, E. Y., & Cheng, R. S., 2007). The degradation profiles of native t-CNC and t-300 CNC-G4 were similar with a slight difference in the temperature that degradation started. For 301 native t-CNC, thermal degradation started at 160 °C whereas for t-CNC-G4 degradation began at 302 215 °C, while it ended at 360 °C for both, meaning that the degradation of t-CNC-G4 is faster 303 (confirmed by a much more intense peak for the differential thermogravimetric (DTG) curve). In 304 the DTG curve, represented by the weight derivative (%/°C), both native t-CNC and t-CNC-G4 305 showed prominent pyrolysis process with two step degradation processes at 295 and 495 °C for 306 native t-CNC and 293 and 480 °C for t-CNC-G4. The slight but significant differences in the 307 thermal stability profile when the dendrimers were synthesized at the reducing ends of t-CNC 308 could be due to the dendrimer functionalization.



Fig. 5. Thermal gravimetric analysis (TGA) and first derivative of the TGA (differential
 thermogravimetric, DTG) curves of native t-CNC and t-CNC-G4.

312

#### 313 *3.2. Reducing-end driven supramolecular assembly*

314 The affinity of amine groups for gold was used to demonstrate the dendrimer growth at the 315 reducing ends of t-CNCs. Hence, the amine-terminated t-CNC-G0, t-CNC-G2 and t-CNC-G4 316 were adsorbed onto gold surfaces, and the adsorption process was monitored by surface plasmon 317 resonance (SPR). SPR is an optical technique based on the resonant oscillation of conduction 318 electrons at the interface between the gold surface and the t-CNC adsorbed layer when 319 stimulated by incident light. Hence, when the gold surface is modified by t-CNC adsorption, the 320 angle at which the plasmon is excited  $(\Delta \theta)$  changes (Ahola, S., Myllytie, P., Osterberg, M., 321 Teerinen, T., & Laine, J., 2008). The adsorbed mass can be calculated from the SPR angle shift, 322 according to Equation 1. Fig. 6 shows the adsorbed mass as a function of time for the adsorption 323 of native t-CNC and the amine-terminated dendrimers t-CNC-G0, t-CNC-G2 and t-CNC-G4 324 onto gold surfaces.



Fig. 6. Calculated surface coverage ( $\Gamma_{SPR}$ ) from SPR angle shift as a function of time showing the adsorption of native t-CNC and the amine-terminated dendrimers t-CNC-G0, t-CNC-G2 and t-CNC-G4. The arrow indicates the injection and the asterisk the rinsing step. The inset depicts the relationship between surface coverage and the number of amine groups on the dendrimer.

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331 The injection of native t-CNC resulted in a slight increase of the adsorbed mass, which indicated 332 some unspecific and very limited adsorption onto the gold surface. The first generation t-CNC-333 G0 showed a similar profile to unmodified t-CNC but the surface coverage was higher. t-CNC-334 G0 only contained one amine group at the reducing end; therefore, the SPR profile could be due 335 to a very slight adsorption on the gold surface. For t-CNC-G2, after injection, we observed a 336 deep increase in the angle shift and therefore the adsorbed mass, even if at the end of the 337 adsorption process, the surface coverage was similar to that of t-CNC-GO. Finally, the last 338 generation t-CNC-G4 showed the largest increase in mass after injection that continued 339 increasing towards a greater adsorbed mass. In all cases, the rinsing step resulted in a slight 340 decrease of the adsorbed mass, indicating low t-CNC-Gx desorption and demonstrating the 341 stability of the adsorbed nanoparticles. The SPR results clearly demonstrated the dendrimer 342 generation growth. Table 2 shows the surface coverage at the end of the adsorption process. In 343 previous studies, we had demonstrated that the adsorption of reducing end-modified t-CNC was 344 impacted by the probability of the nanocrystal to have the proper orientation to adsorb (Chemin, 345 M. et al., 2020). Hence, reducing end-modified t-CNC should be rather perpendicular to the 346 surface, with the reducing end close enough to the surface to facilitate interactions and the 347 subsequent adsorption. The dendrimer generation growth increased the number of amino groups 348 at the t-CNC reducing ends, increasing the size and therefore boosting the probability to 349 encounter the surface. This assumption was demonstrated by the linear relationship between the 350 number of amino groups and the surface coverage calculated by SPR (inset of Fig. 6).

351 In order to get more insight into the arrangement of amine-terminated t-CNC-Gx on the gold 352 substrate, their adsorption was monitored by quartz crystal microbalance with dissipation (QCM-353 D). The evolution of frequency  $(\Delta f_n/n)$  and dissipation  $(\Delta D_n)$  signals for the overtone number n =354 5 as a function of time for t-CNC-G0, t-CNC-G2 and t-CNC-G4 dispersions are shown in Fig. 7a 355 and b. In all cases, the frequency decrease upon t-CNC-Gx injection was very low (from -1 to -3 356 Hz), indicating a very small t-CNC-Gx mass adsorbed on the gold substrate (Moreau, C., Beury, 357 N., Delorme, N., & Cathala, B., 2012). In previous studies, we have already shown that 358 unmodified CNC did not adsorb on gold substrates (Chemin, Maud et al., 2020; Villares, A. et 359 al., 2018), which showed that the adsorption was driven by the functionalities at the reducing 360 ends. The frequency decrease reflected the t-CNC-Gx adsorbed mass together with the hydrating 361 water. Therefore, the surface coverage calculated from QCM-D by the Sauerbrey's expression 362 (equation 2) was in all cases higher than those calculated from SPR data (Table 2).

#### **Table 2** 364

Surface coverage calculated from SPR measurements ( $\Gamma_{SPR}$ ) and QCM-D frequency ( $\Gamma_{QCM}$ ) for the amine-terminated dendrimers t-CNC-G0, t-CNC-G2 and t-CNC-G4. Results are expressed as mean  $\pm$  standard deviation

	$\Gamma_{SPR} (\mathrm{mg \ m}^{-2})$	$\Gamma_{QCM} (\mathrm{mg \ m}^{-2})$
t-CNC-G0	$0.046\pm0.026$	$0.296\pm0.056$
t-CNC-G2	$0.047\pm0.020$	$0.099\pm0.054$
t-CNC-G4	$0.106\pm0.036$	$0.187 \pm 0.159$

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369 QCM-D allows monitoring dissipation changes concomitantly with frequency variations. Hence, 370 the different viscoelastic nature of the adsorbed t-CNC-Gx nanoparticles was investigated by the 371 change of dissipation as a function of frequency ( $\Delta D_n vs \Delta f_n/n$ ). Fig. 7c shows the  $\Delta D_n vs \Delta f_n/n$ 372 plots for the adsorption of t-CNC-G0, -G2 and -G4. Usually, a small slope value of the  $\Delta D_n vs$ 373  $\Delta f_n/n$  curve indicates a relatively thin and rigid adsorbed layer whereas high slope values are 374 characteristic of thicker and viscoelastic layers (Kishani, S., Vilaplana, F., Ruda, M., Hansson, 375 P., & Wagberg, L., 2020; Rodahl, M. et al., 1997).



Fig. 7. Frequency  $(\Delta f_n/n)$  and dissipation  $(\Delta D_n)$  changes for the overtone n = 5 monitored by QCM-D during adsorption of t-CNC-G0, t-CNC-G2 and t-CNC-G4: (a) frequency as a function of time; (b) dissipation as a function of time; and (c)  $\Delta D_n vs \Delta f_n/n$  plots. The arrow indicates the injection of t-CNCs and the asterisk the rinsing step.

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The slopes values were 0.09, 0.33 and  $0.49 \cdot 10^{-6} \text{ Hz}^{-1}$  for t-CNC-G0, t-CNC-G2 and t-CNC-G4, 382 383 respectively. The increase in the slope values as the generation grew indicated higher 384 viscoelasticity of the anchoring group for higher dendrimer generations. This illustrated that the 385 dendrimer size and the number of peripheral amino groups at the reducing ends of t-CNC 386 determined the viscoelastic properties and therefore the interaction between t-CNC-Gx and the 387 surface. Hence, as the generation increased, the t-CNC-Gx could have more freedom to form 388 more flexible layers. It has been previously postulated that the reducing end-modified CNC 389 adsorb on the gold surface by their functionality at the reducing end, and repel each other 390 electrostatically due to the sulfate groups on the CNC surface, resulting in a rather upright 391 orientation (Chemin, Maud et al., 2020; Lokanathan, A. R. et al., 2013; Villares, A. et al., 2018). 392 In previous studies on the "grafting onto" of PAMAM dendrimers on CNC reducing ends, we 393 have observed an increase in the  $\Delta D_n$  vs  $\Delta f_n/n$  slope as the dendrimer generation increased 394 (Chemin, Maud et al., 2020). In this study, we obtained similar slope values, and the gradual 395 increase observed for t-CNC-G0 to t-CNC-G4 could point at a different adsorption regimes as 396 the generation increased. Indeed, the generation growth results in a higher number of amino 397 groups together with an increase of the dendrimer volume and flexibility compared to the rigid 398 nanocrystal. This was reflected by the increasing dissipation values, which was a supplementary 399 proof to demonstrate the dendrimer growth at the reducing end.

#### 401 **4. Discussion**

402 The chemical polarity of cellulose nanocrystals has motivated the selective introduction of 403 functionalities at only one end of the rod-like CNCs (Lin, F. et al., 2019; Lokanathan, A. R. et 404 al., 2013; Villares, A. et al., 2018). The high reactivity of the aldehyde provides a specific 405 anchoring point for chemical synthesis. Generally, macromolecules are introduced by "grafting 406 onto" reactions, which involve coupling reactions between the terminal aldehyde and the 407 targeted polymer (Chemin, Maud et al., 2020; Lin, F. et al., 2019; Sipahi-Saglam, E., Gelbrich, 408 M., & Gruber, E., 2003; Tang, C. X. et al., 2018; Tao, H., Dufresne, A., & Lin, N., 2019; 409 Tavakolian, M. et al., 2019). Differently, the "grafting from" approach includes the introduction 410 of initiator molecules to carry out polymerization. Despite the advantages in terms of 411 controllable polymerization and versatility of available monomers, the "grafting from" strategy 412 has been scarcely used for polymer growth at the CNC reducing end. Zoppe et al. described for 413 the first time the polymers synthesis at the reducing ends of CNC (Zoppe, J. O. et al., 2017). The 414 introduction of initiators at reducing ends and the subsequent addition of the monomers N-415 isopropylacrylamide [2-(methacryloyloxy)ethyl]-trimethylammonium (NIPAM), chloride 416 (METAC) and sodium 4-vinylbenzenesulfonate (4-SS) yielded nanorods with one end-tethered 417 polymer chains by atom transfer radical polymerization (ATRP). By a similar approach, the 418 same group described the polymerization of the thermoresponsive poly(*N*-isopropylacrylamide) 419 (PNIPAM) at the reducing ends of CNC (Risteen, B. et al., 2018).

420 In this work, we describe for the first time the *in situ* growth of poly(amidoamine) dendrimers at 421 the reducing end of cellulose nanocrystals. Our synthetic approach involved straightforward 422 reactions in aqueous media and purification was easily achieved by phase separation. The growth 423 of dendrimer generations was easily controlled by UV and FTIR spectroscopies, and we 424 successfully introduced 4 generations. The better resolution of the FTIR band at 1720 cm<sup>-1</sup> for 425 the acid-terminated generations permitted to distinguish between the peripheral groups, amino or 426 carboxylic acid, of each generation.

427 The divergent approach has been previously used for grafting dendrimers onto the whole CNC 428 surface. Hence, poly(propylene imine) (PPI) and poly(amidoamine) (PAMAM) have been 429 prepared up to the fourth generation. CNC were firstly aminated and dendrimer growth was 430 performed by iterative Michael additions (Golshan, M. et al., 2017; Hassan, M. L., 2006; 431 Tehrani, A. D. & Basiryan, A., 2015; Wang, Y. C. & Lu, Q. Y., 2020). The grafting density in terms of amino groups was 20.1 mmol g<sup>-1</sup> for the fourth PPI generation (Golshan, M. et al., 432 2017) and 3.54 mmol g<sup>-1</sup> for the fourth PAMAM generation (Wang, Y. C. & Lu, Q. Y., 2020), 433 434 both at the CNC surface. In both cases, the coverage of the CNC surface by the dendrimer 435 moieties was high, which resulted in CNC aggregation (Golshan, M. et al., 2017; Wang, Y. C. & 436 Lu, Q. Y., 2020). The advantage of growing dendrimers specifically at the reducing end is the 437 possibility of preserving the CNC surface properties but introducing new functionalities at only 438 one end. Furthermore, this patchy functionalization allows the supramolecular assembly of CNC by their reducing end, even if the grafting density was significantly lower (0.14 mmol g<sup>-1</sup>), 439 440 resulting in hairy layers (Lokanathan, A. R. et al., 2013). Hence, this work demonstrates that the 441 reducing end dendrimer-grafting allowed the adsorption of t-CNC-Gx onto gold surfaces by their 442 modified reducing end. This is of special importance for designing novel materials by 443 supramolecular assembly where CNC can be specifically bound by one end.

444

#### 445 **5.** Conclusions

446 This work describes a straightforward "grafting onto" strategy to synthesize poly(amidoamine) 447 dendrimers at the reducing ends of tunicate cellulose nanocrystals. The generation growth was 448 monitored by UV and FTIR spectroscopies. This work demonstrates that SPR and QCM-D are 449 suitable techniques for detecting functionalities at the nanocrystal reducing ends, and the  $\Delta D_n vs$ 450  $\Delta f_n/n$  plots allow to discern the arrangement of nanocrystals at the gold surface. Concluding, 451 cellulose nanocrystals can be viewed as a good strategy for the step-by-step synthesis of 452 polymers, especially dendrimers since they act as a solid support that facilitates dendrimer 453 purification. Indeed, the presence of dendritic structures at the reducing ends dictates their 454 interaction with gold surfaces, and increase the versatility of cellulose materials.

455

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462

- 463 *Declaration of competing interest*
- 464 The authors declare no conflicts of interest.

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#### 466 **References**

Ahola, S., Myllytie, P., Osterberg, M., Teerinen, T., & Laine, J. (2008). Effect of polymer
adsorption on cellulose nanofibril water binding capacity and aggregation. *Bioresources*,
3(4), 1315-1328.

- Benkaddour, A., Journoux-Lapp, C., Jradi, K., Robert, S., & Daneault, C. (2014). Study of the
  hydrophobization of TEMPO-oxidized cellulose gel through two routes: amidation and
  esterification process. *Journal of Materials Science*, 49, 2832-2843.
- 473 Charreau, H., Cavallo, E., & Foresti, M. L. (2020). Patents involving nanocellulose: Analysis of
  474 their evolution since 2010. *Carbohydrate Polymers*, 237.
- Chemin, M., Moreau, C., Cathala, B., & Villares, A. (2020). Adsorption Behavior of Reducing
  End-Modified Cellulose Nanocrystals: A Kinetic Study Using Quartz Crystal
  Microbalance. *Journal of Renewable Materials*, 8(1), 29-43.
- Chemin, M., Moreau, C., Cathala, B., & Villares, A. (2020). Asymmetric modification of
  cellulose nanocrystals with PAMAM dendrimers for the preparation of pH-responsive
  hairy surfaces. *Carbohydrate Polymers*, 249, 116779.
- 481 Delepierre, G., Heise, K., Malinen, K., Koso, T., Pitkänen, L., Cranston, E. D., ... King, A. W.
  482 T. (2021). Challenges in Synthesis and Analysis of Asymmetrically Grafted Cellulose
  483 Nanocrystals via Atom Transfer Radical Polymerization. *Biomacromolecules*, 22(6),
  484 2702-2717.
- 485 Eichhorn, S. J. (2011). Cellulose nanowhiskers: promising materials for advanced applications.
   486 Soft Matter, 7(2), 303-315.
- 487 Golshan, M., Salami-Kalajahi, M., Roghani-Mamaqani, H., & Mohammadi, M. (2017).
   488 Poly(propylene imine) dendrimer-grafted nanocrystalline cellulose: Doxorubicin loading 489 and release behavior. *Polymer*, 117, 287-294.
- Guo, Y. Q., Shen, M. W., & Shi, X. Y. Construction of Poly(amidoamine) Dendrimer/Carbon
   Dot Nanohybrids for Biomedical Applications. *Macromolecular Bioscience*.
- Hassan, M. L. (2006). Preparation and thermal stability of new cellulose-based poly(propylene
   imine) and poly(amido amine) hyperbranched derivatives. *Journal of Applied Polymer Science*, 101(3), 2079-2087.
- Heise, K., Delepierre, G., King, A., Kostiainen, M., Zoppe, J., Weder, C., & Kontturi, E. (2020).
  Chemical modification of cellulose nanocrystal reducing end-groups. *Angewandte Chemie International Edition*, 60(1), 66-87.
- Heise, K., Kontturi, E., Allahverdiyeva, Y., Tammelin, T., Linder, M. B., Nonappa, & Ikkala, O.
  (2020). Nanocellulose: Recent Fundamental Advances and Emerging Biological and Biomimicking Applications. *Advanced Materials*, 33(3), 2004349.
- Kishani, S., Vilaplana, F., Ruda, M., Hansson, P., & Wagberg, L. (2020). Influence of Solubility
   on the Adsorption of Different Xyloglucan Fractions at Cellulose-Water Interfaces.
   *Biomacromolecules*, 21(2), 772-782.
- Kontturi, K. S., Kontturi, E., & Laine, J. (2013). Specific water uptake of thin films from
   nanofibrilar cellulose. *Journal of Materials Chemistry A*, 1, 13655-13663.
- Lin, F., Cousin, F., Putaux, J.-L., & Jean, B. (2019). Temperature-Controlled Star-Shaped
   Cellulose Nanocrystal Assemblies Resulting from Asymmetric Polymer Grafting. Acs
   Macro Letters, 8, 345-351.
- Lin, N., & Dufresne, A. (2014). Surface chemistry, morphological analysis and properties of
   cellulose nanocrystals with gradiented sulfation degrees. *Nanoscale*, 6(10), 5384-5393.
- Lin, N., Huang, J., Chang, P. R., Anderson, D. P., & Yu, J. (2011). Preparation, Modification,
  and Application of Starch Nanocrystals in Nanomaterials: A Review. *Journal of Nanomaterials*.

- Lokanathan, A. R., Lundahl, M., Rojas, O. J., & Laine, J. (2014). Asymmetric cellulose
   nanocrystals: thiolation of reducing end groups via NHS-EDC coupling. *Cellulose*, 21(6),
   4209-4218.
- Lokanathan, A. R., Nykanen, A., Seitsonen, J., Johansson, L.-S., Campbell, J., Rojas, O. J., ...
   Laine, J. (2013). Cilia-Mimetic Hairy Surfaces Based on End-Immobilized Nanocellulose
   Colloidal Rods. *Biomacromolecules*, 14(8), 2807-2813.
- Moreau, C., Beury, N., Delorme, N., & Cathala, B. (2012). Tuning the architecture of cellulose
   nanocrystal-poly(allylamine hydrochloride) multilayered thin films: Influence of dipping
   parameters. *Langmuir*, 28(28), 10425-10436.
- Nikzamir, M., Hanifehpour, Y., Akbarzadeh, A., & Panahi, Y. Applications of Dendrimers in
   Nanomedicine and Drug Delivery: A Review. *Journal of Inorganic and Organometallic Polymers and Materials.*
- Potthast, A., Radosta, S., Saake, B., Lebioda, S., Heinze, T., Henniges, U., . . . Wetzel, H.
  (2015). Comparison testing of methods for gel permeation chromatography of cellulose:
  coming closer to a standard protocol. *Cellulose*, 22(3), 1591-1613.
- Risteen, B., Delepierre, G., Srinivasarao, M., Weder, C., Russo, P., Reichmanis, E., & Zoppe, J.
  (2018). Thermally Switchable Liquid Crystals Based on Cellulose Nanocrystals with
  Patchy Polymer Grafts. *Small*, 14(46).
- Rodahl, M., Hook, F., Fredriksson, C., Keller, C. A., Krozer, A., Brzezinski, P., . . . Kasemo, B.
  (1997). Simultaneous frequency and dissipation factor QCM measurements of
  biomolecular adsorption and cell adhesion. *Faraday Discussions*, 107, 229-246.
- Sacui, I. A., Nieuwendaal, R. C., Burnett, D. J., Stranick, S. J., Jorfi, M., Weder, C., ... Gilman,
  J. W. (2014). Comparison of the Properties of Cellulose Nanocrystals and Cellulose
  Nanofibrils Isolated from Bacteria, Tunicate, and Wood Processed Using Acid,
  Enzymatic, Mechanical, and Oxidative Methods. Acs Applied Materials & Interfaces,
  6(9), 6127-6138.
- Sauerbrey, G. (1959). Verwendung von Schwingquarzen zur Wägung dünner Schichten und zur
   Mikrowägung. *Zeitschrift für Physik*, 155(2), 206-222.
- Schoch, R. L., & Lim, R. Y. H. (2013). Non-Interacting Molecules as Innate Structural Probes in
  Surface Plasmon Resonance. *Langmuir*, 29(12), 4068-4076.
- Schwanninger, M., Rodrigues, J. C., Pereira, H., & Hinterstoisser, B. (2004). Effects of short time vibratory ball milling on the shape of FT-IR spectra of wood and cellulose.
   *Vibrational Spectroscopy*, 36(1), 23-40.
- Shea, F., & Watts, C. E. (1939). Dumas method for organic nitrogen. *Industrial and Engineering Chemistry Analytical Edition*, 11(6), 333-334.
- Sipahi-Saglam, E., Gelbrich, M., & Gruber, E. (2003). Topochemically modified cellulose.
   *Cellulose*, 10(3), 237-250.
- Tang, C. X., Spinney, S., Shi, Z. Q., Tang, J. T., Peng, B. L., Luo, J. H., & Tam, K. C. (2018).
   Amphiphilic Cellulose Nanocrystals for Enhanced Pickering Emulsion Stabilization.
   *Langmuir*, 34(43), 12897-12905.
- Tao, H., Dufresne, A., & Lin, N. (2019). Double-Network Formation and Mechanical
  Enhancement of Reducing End-Modified Cellulose Nanocrystals to the Thermoplastic
  Elastomer Based on Click Reaction and Bulk Cross-Linking. *Macromolecules*, 52(15),
  5894-5906.

- Tao, H., Lavoine, N., Jiang, F., Tang, J., & Lin, N. (2020). Reducing end modification on
   cellulose nanocrystals: strategy, characterization, applications and challenges. *Nanoscale Horizons*, 5(4), 607-627.
- Tavakolian, M., Lerner, J., Tovar, F. M., Frances, J., van de Ven, T. G. M., & Kakkar, A. (2019).
   Dendrimer directed assembly of dicarboxylated hairy nanocellulose. *Journal of Colloid* and Interface Science, 541, 444-453.
- Tehrani, A. D., & Basiryan, A. (2015). Dendronization of cellulose nanowhisker with cationic
   hyperbranched dendritic polyamidoamine. *Carbohydrate Polymers*, 120, 46-52.
- Terech, P., Chazeau, L., & Cavaille, J. Y. (1999). A small-angle scattering study of cellulose
  whiskers in aqueous suspensions. *Macromolecules*, 32(6), 1872-1875.
- Villares, A., Moreau, C., & Cathala, B. (2018). Star-like Supramolecular Complexes of
   Reducing-End-Functionalized Cellulose Nanocrystals. *Acs Omega*, 3(11), 16203-16211.
- Wang, N., Ding, E. Y., & Cheng, R. S. (2007). Thermal degradation behaviors of spherical
  cellulose nanocrystals with sulfate groups. *Polymer*, 48(12), 3486-3493.
- Wang, Y. C., & Lu, Q. Y. (2020). Dendrimer functionalized nanocrystalline cellulose for Cu(II)
   removal. *Cellulose*, 27(4), 2173-2187.
- Zhao, Y., & Li, J. (2014). Excellent chemical and material cellulose from tunicates: diversity in
   cellulose production yield and chemical and morphological structures from different
   tunicate species. *Cellulose*, 21(5), 3427-3441.
- Zhao, Y. D., Zhang, Y. J., Lindstrom, M. E., & Li, J. B. (2015). Tunicate cellulose nanocrystals:
   Preparation, neat films and nanocomposite films with glucomannans. *Carbohydrate Polymers*, 117, 286-296.
- Zoppe, J. O., Dupire, A. V. M., Lachat, T. G. G., Lemal, P., Rodriguez-Lorenzo, L., Petri-Fink,
  A., . . Klok, H.-A. (2017). Cellulose Nanocrystals with Tethered Polymer Chains:
  Chemically Patchy versus Uniform Decoration. *Acs Macro Letters*, 6(9), 892-897.