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1	Multiscale investigation of bonded granular materials: the H-bond
2	model

Zeyong Liu<sup>a</sup>, Francois Nicot<sup>a</sup>, Antoine Wautier<sup>b</sup>, Felix Darve<sup>c</sup>
 <sup>a</sup> Université Savoie Mont-Blanc, ISTerre, Le Bourget-du-Lac, France
 <sup>b</sup> Aix-Marseille University, INRAE, RECOVER, Aix-en-Provence, France
 <sup>c</sup> Université Grenoble-Alpes, Laboratoire Sols Solides Structures, Grenoble, France

# 7 Abstract

Cemented granular materials play an important role in both natural and engineered 8 structures, as they are able to resist traction forces. However, modeling the mechanical 9 10 behavior of such materials is still challenging, and most of existing constitutive models follow phenomenological approaches that unavoidably disregard the microstructural 11 12 mechanisms taking place on the bonded grains scale. This paper presents a multiscale approach applicable to any kind of granular materials with solid bonds between 13 particles. Inspired from the H-model, this approach allows simulating the behavior of 14 cemented materials along various loading paths, by describing the elementary 15 mechanisms taking place between bonded grains. In particular, the effect of local bond 16 failure process on the macroscopic response of the whole specimen is investigated 17 according to the bond strength characteristics. 18

Key words: Cemented granular materials, bonded contact model, H-model, meso structure, multiscale approach, homogenization

### 22 **1. Introduction**

Cohesive geomaterials are widespread in a variety of engineering purposes (such as 23 natural and artificial cemented sands, concrete, and sedimentary rocks), where solid 24 bonding is known as an important characteristic (Pettijohn et al., 1987; Leroueil and 25 26 Vaughan, 1990; Cuccovillo and Coop, 1999). The solid bonding includes i) natural cementation originating from various processes and infill materials such as oxidative 27 precipitates or clays (Cuccovillo and Coop, 1997; Ismail et al., 2002; Yin and Karstunen, 28 2011; Lin et al., 2016) and ii) artificial cementation where soil can be mixed with 29 cement, lime, bacteria producing calcite or other adhesive materials (Lade and Overton, 30 1989; Huang and Airey, 1998; Gao and Zhao, 2012; Rios et al., 2014; Montoya and 31 DeJong, 2015; Terzis and Laloui, 2018; Nafisi et al., 2019; Xiao et al., 2021). Through 32 the influence of solid bonds, these cemented geomaterials typically exhibit distinctive 33 behaviors in comparison to their unbonded counterparts, and it has been recognized that 34 they play an important role in engineering (Leroueil and Vaughan, 1990; Kochmanová 35 and Tanaka, 2011; Rahman et al., 2010). Hence, solid bonds should be taken into 36 account to better understand the mechanical behavior of cemented geomaterials. 37

38 Historically, experiments that capture the mechanical behavior of solid bonds have shown that many peculiar features distinguish cemented geomaterials from non-39 cemented ones such as a substantial softening for stress-strain response, a slight 40 increase in residual friction at critical state (Rahman et al., 2018) and a more dilative 41 volumetric response (Abdulla and Kiousis, 1997; Tang et al., 2007; Feng et al., 2017). 42 This was put forward for both natural (Burland, 1990; Rouainia and Muir wood, 2000; 43 44 Rocchi et al., 2003) or artificial cemented specimens (Coop and Atkinson, 1994; 45 Consoli et al., 2007; Gao and Zhao, 2012), which motivated the development of several constitutive models for cemented geomaterials (Rotta et al., 2003; Rabbi et al., 2011; 46 Jiang et al., 2013). 47

As it remains challenging to observe and quantify the failure of cementation at 48 microscale, a global phenomenological approach is often used in constitutive modeling 49 50 (Rouainia and Muir wood, 2000; Taheri et al., 2012). However, such models fail in 51 properly accounting for the underpinning microscopic mechanisms. As an alternative, the discrete element method (DEM) emerges as a potent tool for enhancing the 52 comprehension of the link between local bond breakage and global, constitutive 53 characteristics in cemented geomaterials (Jiang et al., 2014a; Wu et al., 2021). This is 54 attributed to its capability to relate macroscopic responses to microscopic information, 55 such as bond breakage (Wang and Leung, 2008). 56

To feed DEM approaches, many previous studies have so far focused on bonded contact models between grains (Obermayr et al., 2013; Jiang et al., 2014a; Shen et al., 2016; Zhang and Dieudonné, 2023). For existing bonded contact models, the bond geometry between two grains is simplified, usually idealizing a bond as a short beam connecting the surfaces of two adjoining grains (Jiang et al., 2014b; Brendel et al., 2011; Yang et al., 2019). Such a microstructure where two particles and a bond are linked together can be referred to as a triad.

64 Previous researchers have developed several continuum constitutive models to

65 describe some important features of cemented granular materials (Kavvadas and Amorosi, 2000; Rocchi et al., 2003; Evans et al., 2014; Li et al., 2017; Khoubani, 2018). 66 Multiscale approaches can be used to describe the bond behavior on the grain scale via 67 specific bonded contact laws. Thanks to homogenization techniques, the macroscopic 68 constitutive properties emerge from the collective response of the microstructural 69 bodies (Mehrabadi et al., 1997; Balendran and Nemat-Nasser, 1993a and 1993b; 70 Nemat-Nasser, 2000; Nemat-Nasser and Zhang, 2002). The micro-directional model 71 72 (Nicot and Darve, 2005) stands as an example of micromechanically-based model, where the granular assembly can be described as a collection of contacts between pairs 73 74 of adjoining spherical grains with different orientations in the physical space. In order 75 to enrich the microstructural description, this model was extended to the H-model (Nicot and Darve, 2011a and 2011b; Xiong et al., 2017; Xiong et al., 2021) by including 76 an intermediate scale. The 2D H-model was derived from hexagonal mesostructures 77 composed of six monodisperse, spherical grains in contact, forming a closed hexagonal 78 79 loop. The H-model was shown to be a very potent micromechanical model in order to 80 capture most of the salient constitutive features of granular materials (Wautier et al., 2021). 81

This paper aims to investigate the influence of solid bonds on the response of 82 granular assemblies, by accounting for underpinning microstructural mechanisms. For 83 this purpose, the 2D H-model is considered and extended by including solid bonds 84 between the grains in contact. The response of different specimens is then analyzed 85 86 along classical loading paths (i.e. biaxial loading path, and proportional strain loading path). Several key aspects are investigated, such as bonded and unbonded global 87 behavior, bond failure mechanisms along the different loading paths. The results are 88 eventually discussed in terms of the macroscopic responses in relation to 89 micromechanical aspects, including detailed analyses on the influences of the bond 90 91 breakage.

92 Time differential of any variable  $\psi$  will be denoted  $\delta \psi$ , as the product of the

93 particulate derivative  $\dot{\psi}$  by the infinitesimal time increment  $\delta t$ .

# 94 2. *H*-model and physical mechanisms of bonded materials

95 2.1 *H*-model in brief

96 The general principle of the *H*-model consists in a statistical description of the 97 microstructure of granular assemblies. A homogenization process is developed by 98 averaging the local behavior taking place at the intermediate scale corresponding to 99 hexagonal sets of grains.

Basically, the elementary unit for the 2D original *H*-model is a hexagonal pattern (denoted H-cell) composed of six spherical particles with the same radius interacting through contact laws. Each cell is assumed to be loaded by a symmetric set of forces, as illustrated in Fig. 1. 104 Compared with the original H-model that accounts for elasto-frictional behavior at contacts, a cemented assembly can be described by updating the contact law. Bond 1 105 106 acts on the contact between particle 1 and particle 2, where both normal and tangential 107 contact forces are present. Bond 2 acts on the contact between particle 2 and particle 3, 108 where only a normal contact force is involved because of the symmetry preservation, as shown in Fig. 1. In the H-model, grains are assumed not to rotate. This is imposed 109 through additional external forces  $G_2$  that ensure the momentum balance for all grains. 110 The testing procedure will be described in detail in section 3, where some classical 111 loading paths involving traction, compression, and shear mechanisms are considered. 112





Fig. 1. Description of forces and bonds in a given hexagonal meso-structure.



115

116 Fig. 2. Global coordinates  $(e_1, e_2)$  and mesoscale coordinates (n, t) (left); geometrical settings 117 (right).

118 2.2 Bonded contact model

Bonded contact models can be presented in the general framework of bonded granular materials (Ismail et al., 2002; Jiang et al., 2014b). The material constituting the bonds between particle can be regarded as a brittle elastic medium. The bonded contact model between grains considers that the two grains and the bond (elementary triad) are deformable. In the elastic regime, as seen in Fig. 3, a triad can be described by three springs mounted in series, both along normal and tangential directions.

125 The contact forces acting at inter-particle contacts (normal contact force  $F_n$  and 126 tangential contact force  $F_t$ ) can be related to the relative displacements (normal 127 component  $u_n$  and tangential component  $u_t$ ) as follows in the elastic regime:

128 
$$\delta F_n = k_n \delta u_n \tag{1}$$

129 
$$\delta F_t = k_t \delta u_t \tag{2}$$

For a serial bonded contact model, the bond material acts as an additional spring in series. The stiffness of the spring triad (Fig. 3) is denoted as  $k_n$  and  $k_t$  for normal direction, and tangential direction, respectively. In a serial assembly, the contact stiffness is controlled by the particle stiffness and bond stiffness as follows:

134 
$$k_n = \frac{k_{np}k_{nb}}{2k_{nb} + k_{np}}$$
 (3)

135 
$$k_{t} = \frac{k_{tp}k_{tb}}{2k_{tb} + k_{tp}}$$
(4)

136 where  $k_{np}$  and  $k_{tp}$  refer to particle stiffness in normal and tangential directions, 137 whereas  $k_{nb}$  and  $k_{tb}$  refer to bond material stiffness in normal and tangential 138 directions.

In this manuscript, once the strength limit of the bond is reached, the bond fails and is considered not to exist any longer. Fig. 4 presents the unbonded contact model after bond failure, in which only two particles contact. In the normal direction, an elastic behavior is considered, and only a compressive force can be transmitted. In the tangential direction, an elasto-frictional law is activated. Details of the classical contact law for the original *H*-model can be found in (Nicot and Darve, 2011b). The residual contact law is expressed as follows:

146 
$$\delta F_n = k'_n \delta u_n \tag{5}$$

(6)

147 
$$\delta F_t = \begin{cases} k_t' \delta u_t & \text{elastic regime} \\ \tan \varphi_g (F_n + k_n' \delta u_n) - F_t & \text{plastic regime} \end{cases}$$

148

For the unbonded contact model, the deformability of particles is modeled through the overlap due to the relative displacement of the two particles. Compared with the bonded contact, once the bond has failed, both normal and tangential stiffnesses of a bond are supposed to be infinite  $(k_{nb} = \infty, k_{tb} = \infty)$ . Thus, the contact stiffness can simply be obtained from equations (3) and (4), and reads:

$$154 k_n' = \frac{k_{np}}{2} (7)$$

 $k_t' = \frac{k_{tp}}{2}$ 



(8)

156

Fig. 3. Serial bonded contact model with mechanical components (orange: mechanical spring for
 particles; black: mechanical spring for bond material).





Fig. 4. Details of the unbonded contact model in the elastic regime.

#### 161 2.3 Bond failure criterion

In order to simplify the constitutive model, it is assumed that a bond between particles breaks abruptly according to a brittle failure type. The bond failure criterion includes three basic modes (compression, traction, and shear), as shown in Fig. 5. Compression failure occurs when the normal contact force exceeds the compressive bond strength. Traction failure occurs when a bond is under traction with a normal contact force reaching the traction bond strength. The limit of normal contact forces can be expressed as:

169

$$F_n = \begin{cases} R_{n,c} & \text{compression} \\ R_{n,t} & \text{traction} \end{cases}$$
(9)

171

170

172 where  $R_{n,c}$  is the compressive bond strength;  $R_{n,t}$  is the traction bond strength. 173 Furthermore, shear failure occurs once the tangential contact force reaches the shear 174 bond strength  $R_s$ :

$$175 F_t = R_s (10)$$

176

Finally, as shown in Fig. 5, three failure mechanisms coexist for bonded contacts. The sign of the normal contact force determines the mechanical regime (compression regime or traction regime, with compressive forces counted positive):

181 
$$\begin{cases} R_{n,c} > F_n > 0 & \text{compression} \\ R_{n,t} < F_n < 0 & \text{traction} \\ |F_t| > R_s & \text{shear} \end{cases}$$
(11)

182

The resulting contact after bond breakage behaves as an unbonded contact. It means that the bond material at a serial bonded contact admits an infinite stiffness at the instant of bond failure. The system immediately transforms into particle-to-particle contact, and the contact forces will evolve along the loading path with the contact law given in equations (5) - (6).

The existence of a bond gives rise to a variation in the contact force when failure 188 occurs. Fig. 6 shows the evolution of the normal contact force with the normal 189 displacement. In compression regime, the normal contact force varies linearly with the 190 normal displacement. After bond failure in a compression regime, the contact between 191 the particles still exists as a particle-to-particle contact. In this case, the change in 192 contact stiffness causes the normal contact force to evolve with another slope. In Fig. 193 194 6, it can be seen that a zero force with a non-zero overlapping will be obtained if an 195 unloading is performed after bond failure. This is consistent with the damage mechanics framework (Mazars and Pijaudier-Cabot, 1989; Giry et al., 2011), inducing a change in 196

197 stiffness directed by the vanishing of the broken bond. Further details can be found in 198 appendix A. In traction regime, the normal contact force increases while the bond exists 199 and drops to zero at the moment of bond failure. The existence of contacts between 200 particles for each H-cell after bond failure is checked from the geometrical parameters 201 of the H-cell. The contact 1 is lost when  $d_1 > 2r$ , and the contact 2 is lost when  $d_2 >$ 202 2r, as shown in Fig. 2. If no contacts exist anymore, the meso-stress for such an H-cell 203 drops down to 0.

As for the tangential contact force, an elastic response exists before bond failure 204 occurs. After failure, the contact model transforms into an elasto-frictional model. 205 Finally, as shown in Fig. 7a, the evolution of contact forces can be represented in the 206 plane of normal and tangential forces, where a bond strength box gives the limit of these 207 components. Four typical failure situations can be identified, as depicted in Fig. 7b with 208 the four colored lines. It can be seen that when the bond fails in a traction regime, the 209 contact forces drop to zero once the bond is broken. When the bond fails in a 210 211 compression regime, the contact is described by an elasto-frictional contact law, with a 212 change in the slope of the normal contact force, as seen in Fig. 7b.



213 214

Fig. 5. Three modes of bond failure for the bonded contact model.





Fig. 6. Mechanical model along the normal contact direction.





218

Fig. 7. Failure surface (a) and mechanical behavior of a triad (b) in the contact forces ( $F_n$  and  $F_t$ ) plane.

# 221 3. Numerical inspection of the *H*-bond model capability

The original *H*-model is able to account for geometric changes in the microstructure of granular materials through the deformation of elementary hexagonal patterns at mesoscale. This feature makes the *H*-model able to capture most of the salient constitutive features of granular assemblies, as observed along classical loading paths (Nicot and Darve, 2011b).

In this section, the capability of the *H*-bond model, equipped with the bonded contact model detailed in the previous section, is analyzed.

229 3.1 Typical mechanical response of cemented materials

Figure 8 presents some typical results extracted from the literature, corresponding to 230 deviatoric stress responses during experimental biaxial loading tests with artificially 231 cemented sands (Wang and Leung, 2008), together with the numerical response for the 232 same materials during DEM simulations (Jiang et al., 2013). The cement content of the 233 experimental specimen was determined from the weight ratio, i.e. the weight of the 234 235 cement to the total dry weight of the soil-cement mixture. The bonded loose specimens 236 used during the DEM simulations were prepared homogenously, which required identical bonds to be formed at each contact. It can be noted that the simulated 237 deviatoric stress curves have the same trend as those obtained experimentally. A 238 softening behavior can be observed for cemented samples after a stress peak is reached, 239 while the uncemented specimens experience a monotonous hardening regime. Fig. 8 240 241 also shows that the increasing bond strength contributes to an increase in the peak stress. In the next section, the ability of the H-bond model to reproduce these features is 242 243 explored.





Fig. 8. Stress-strain responses of granular specimens: (a) experimental data (Wang and Leung, 2008);
(b) DEM results (Jiang et al., 2013).

247 3.2 Numerical simulations of cemented materials using the H-bond model

An isotropic statistical distribution of H-cells is used throughout this section.
The general scheme of the *H*-bond-model is summarized as follows:
The strain homogeneity hypothesis allows to update the H-cell geometry in accordance with the macroscopic strain increments.
The relative deformations at contact scale for each H-cell are computed to fulfill static equilibrium.
The contact forces are updated based on the incremental evolution of the H-cell

- 255 geometry.
- 4. The macroscopic stresses are ultimately derived by statistical averaging of all
   meso-stresses acting within each H-cell.
- 258

The numerical parameters used to run the simulation are reported in Table 1. It is worth noting that bonds and particles are supposed to be made up of the same material. The micromechanical parameters are therefore the same for bonds and particles. In the simulations, an isotropic consolidation is first imposed until a confining pressure of 100 kPa is reached. Then, the sample undergoes a given loading path by imposing a strain loading in the axial direction, while specific lateral loading conditions are prescribed.

- 265 Table 1
- 266 <u>Summary of parameters used for the *H*-bond model simulation.</u>

Parameter	Symbol	Unit	Value
Normal contact stiffness of particles	$k_{np}$	N/m	4*10 <sup>8</sup>
Normal contact stiffness of bond	k <sub>nb</sub>	N/m	$4*10^{8}$
Contact stiffness ratio	$\frac{k_t}{k_n}$	-	0.5
Compressive bond strength	R <sub>n,c</sub>	kN	500

Tractive bond strength	$R_{n,t}$	kN	500
Shear bond strength	R <sub>s</sub>	kN	900
Inter-granular friction angle	$arphi_g$	deg	30
Initial opening angle	α <sub>0</sub>	deg	40
Number of directions	$N_{\theta}$	-	360

267 Biaxial loading and proportional strain loading paths are considered in order to combine compression, traction and shear failure regimes at the mesoscopic scale. This 268 allows to mix different bond failure modes along the mechanical response of the 269 specimens, and analyze how they interact with each other. 270

A biaxial test is first simulated under a constant lateral stress, while a constant strain 271 rate is imposed in the axial direction: both  $\dot{\varepsilon}_1$  and  $\sigma_2$  are constant. Proportional strain 272 loading paths are also considered. For such loading paths, the imposed axial strain rate 273 is constant while the lateral stain rate is proportional to the axial strain rate:  $\dot{\varepsilon}_2 = \lambda \dot{\varepsilon}_1$ . 274 As the incremental volumetric strain is given by  $\dot{\varepsilon}_v = (1 + \lambda)\dot{\varepsilon}_1$ , three regimes can be 275 explored: 276

(12)

277

279

 $\begin{cases} \lambda > -1 & \text{contractant regime} \\ \lambda = -1 & \text{isochoric regime} \\ \lambda < -1 & \text{dilatant regime} \end{cases}$ 278

3.2.1 Biaxial loading path

The macroscopic stress and strain responses are explored in 2D conditions by 280 considering the mean stress  $p = (\sigma_1 + \sigma_2)/2$ , the deviatoric stress  $q = \sigma_2 - \sigma_1$ , and 281 the volumetric strain  $\varepsilon_v = \varepsilon_1 + \varepsilon_2$ , where  $\sigma_1$  and  $\sigma_2$  are the principal stresses, and 282  $\varepsilon_1$  and  $\varepsilon_2$  are the principal strains oriented along axial (1) and lateral (2) directions. 283

A dense sample is considered, either with bonded particles, or with unbonded 284 particles. The unbonded specimen is considered as a reference. The typical macroscopic 285 responses obtained are reported in Fig. 9. 286

The stress-strain responses of the cemented specimen differ from the response of the 287 unbonded specimen. It can be observed that the bond effect is dominant in the early 288 stage of loading, where a higher stress peak occurs for the cemented specimen, followed 289 by a more pronounced softening behavior. Likewise, larger contractancy develops at 290 small strains for the cemented specimen. 291



Fig. 9. Deviatoric stress and volumetric strain responses along the biaxial loading; cemented and uncemented specimens.





297

Fig. 10. Deviatoric stress along the biaxial test with different bond strengths; (a) different shear
 strengths, and (b) different compression/traction strengths.

It can also be observed in Fig. 9 that the deviatoric stress curve exhibits two peaks. The first stress peak stems from an intense bond failure activity, occurring along specific orientations of the H-cells. For these cells, the bonded contacts transform into cohesionless contacts. Among them, some H-cells were in traction regime at failure, which results in a contact opening with subsequent macroscopic stress decrease. Ultimately, when most of the bonds have broken, the specimen evolves as the unbonded specimen: a second stress peak is reached, followed by a softening regime.

307 The influence of the bond strength (in traction, compression, or shear) is explored in Fig. 10. For this purpose, different values of bond strength are considered. The first case 308 corresponds to different shear strengths ( $R_{n,c} = R_{n,t} = 500$  kN;  $R_s = 600$  kN, 800 309 kN), as shown in Fig. 10a. It can be observed that the increase in shear strength leads 310 to an increase in the first peak stress. The failure of bond in shear occurs first, and 311 312 affects the response of bond failure in traction due to the change of stiffness. The increase in shear strength delays the bond failure in shear, and likewise delays the bond 313 failure in traction, which results in an increase in the first peak stress. Fig. 10b shows 314 the effects of different traction and compression strengths  $(R_{n,c} = R_{n,t} = 500 \text{ kN},$ 315 800kN;  $R_s = 600$  kN). The traction and compression strengths mainly affect the 316 loading sequence during which the failure in traction occurs successively within the 317 bonds belonging to different cell orientations. Thus, the increase in the traction strength 318 makes the bonded specimen convert into an unbonded specimen  $(R_{n,c} = R_{n,t} = 800 \text{kN};$ 319 320  $R_s = 600$  kN), as shown in Fig. 10b.

### 321 3.2.2 Proportional strain loading path

The evolution of the deviatoric stress as a function of the axial strain is given in Fig. 322 11. Contractant, isochoric, and dilatant proportional strain loading paths are considered 323 with  $\lambda = -0.8, -1, -1.2$ , respectively. For the contractant case ( $\lambda = -0.8$ ), it can be 324 325 observed that the deviatoric stress increases continuously. When  $\lambda = -1$ , zero volume change is imposed, and the deviatoric stress increases until it reaches a plateau. When 326 327  $\lambda = -1.2$ , the deviatoric stress increases at first and then decreases gradually after the peak, corresponding to a static liquefaction process where most of the bonded contacts 328 329 open.

The bond failure evolution is given in Fig. 12. It can be observed that the bonds fail 330 in compression regime first, whatever the volumetric strain regime (contractant, dilatant 331 or isochoric). This is due to the fact that at the beginning of the loading, the bonds with 332 the largest stress are in a compressive regime. This is all the more noticeable when the 333 volumetric strain regime is contractant. Then, the contacts behaving in a traction regime 334 335 fail once the tensile strength is reached. It should be noted that this bond failure evolution depends on the relative bond strengths in compression, tension and shear. For 336 337 example, much larger values in compression strengths could lead to a different scenario by limiting the bond failure in compression. 338 339







345

Fig. 12. Map of bond failure (bond 2) along the three proportional strain loading paths in terms of Hcells distribution. For each H-cell orientation (θ), the axial strain at which bond 2 fails is given.
Failure in compression appears in red while failure in traction appears in blue. The vertical
dotted line highlights the axial strain at which bonds start failing in traction.

350 3.3 Quantitative analysis of bond failure

In order to track the process of bond failure during the loading path, the concepts of bond failure ratio and bond failure rate are introduced. The bond failure ratio is defined as the number of broken bonds over the total number of bonds, and can be expressed as:

$$R_i = \frac{n_b}{N_b} \tag{13}$$

where  $N_b$  is the total number of bonds at the initial state, and  $n_b$  is the cumulated number of broken bonds at a given axial strain  $\varepsilon_1$ . With six bonded contacts per H-cell,  $N_b$  is equal to six times the number of H-cells.

The bond failure rate is then defined as the bond failure ratio increment over a given (constant) axial strain increment, and can be expressed as:

$$r_i = \frac{dR_i}{d\varepsilon_1} \tag{14}$$

where  $dR_i$  is the bond failure ratio increment observed over the axial strain 362 363 increment  $d\varepsilon_1$ .

Figure 13 focuses on the evolution of the bond failure rate with respect to the axial 364 strain in the beginning of the biaxial loading. It reveals that the peak of the bond failure 365 rate coincides with the peak stress, and is followed by a sharp, subsequent decrease. 366 The evolution of the bond failure ratio along the loading path is given in Fig. 14. The 367 figure shows that the bond failure ratio increases with two different regimes. The A-B 368 stage corresponds to a compressive bond failure regime, whereas the B-D stage 369 corresponds to a shear and a traction bond failure regime. 370

Two peaks are observed (Fig. 13b). The first peak (point A) corresponds to the bonds 371 372 undergoing a compressive failure. As a consequence, this first peak has no important 373 effect on the stress-strain response curve of the specimen. The second peak is observed 374 at point B, which nearly corresponds to the peak stress of the deviatoric stress curve. At this stage, the tensile failure of bonds takes place massively, as shown in Fig. 14b. As 375 the bonds break in traction, contacts open, and the corresponding H-cells vanish, 376 resulting in fewer H-cells contributing to the macroscopic stress. Hence, a remarkable 377 reduction in the macroscopic stress occurs, leading to a noticeable softening. 378







382 Fig. 13. Deviatoric stress evolution along a biaxial loading path, with the axial strain up to 4% (a), 383 and corresponding evolution of bond failure rate (b).



385

386 Fig. 14. Evolution of the bond failure ratio with the axial strain (a), and corresponding bond failure modes (b). 387



3.4 Micro-mechanical analysis of failure in cemented materials 388

In order to better understand how the bond failure evolution affects the macroscopic 389 response of a given cemented specimen, the microscopic mechanical responses 390 391 obtained from the H-bond model on the H-cell scale, such as contact forces or mesoscopic strains, are investigated in this section. 392

Fig. 15 shows that the volumetric strain of the cemented specimen gives rise to a 393 strong contractancy followed with a quick volume change at the axial state (0.86%) due 394 to massive bond failure in traction regime (Fig. 14b). The lateral strain  $\varepsilon_2$  of the 395 cemented specimen is first in compression and then gradually evolves into extension 396 397 with the purpose of keeping  $\sigma_2$  constant, which leads to a transition along some H-cell 398 directions from a compression regime to a traction regime. Bond failure in traction regime also has a noticeable effect at the axial state  $\varepsilon_2 = 0.86\%$ . The subsequent 399 contact opening results in a sharp increase in the lateral strain of the cemented specimen, 400 as shown in Fig. 15b. 401



405 Fig. 15. Volumetric strain responses (a) and lateral strain responses (b) along the biaxial loading for





409 Fig. 16. Maps of bond failure occurrence along a biaxial loading path. For each H-cell orientation (θ),
410 the axial strain at which bond 1 fails is given. Failure in compression appears in dark green,
411 failure in shear appears in light blue and failure in traction appears in red. The vertical dotted
412 lines highlight the axial strain at which bonds start failing in shear and in traction.

413

414 Fig. 16 presents the bond failure evolution with respect to the axial strain. Whatever the bond category (bond 1 or bond 2), the bonds break in compression regime first in 415 the early stage of the loading. Then, the bonds 2 break in a traction regime. This is due 416 to the fact that H-cells orientations are distributed in the range from 0 deg to 180 deg 417  $(\theta = 0 \text{ deg corresponds to H-cell aligned with the axial direction, while } \theta = 90 \text{ deg}$ 418 to H-cells aligned with the lateral direction). In terms of H-cell orientations, three 419 420 domains (A, B, and C) are illustrated with respect to different regimes of bond failure, as shown in Fig. 16b. In the domain A, the response of H-cells is dominated by the 421 failure of bond 1 in shear and the failure of bond 2 in traction. In the domain C, both 422 bond 1 and bond 2 experience failure in a compression regime. In the domain B, the 423 failure of bond 1 in compression influences the behavior of bond 2 due to the change 424 in contact stiffness  $(k_n \text{ and } k_t)$ , making bond 2 brake earlier in a traction regime. 425





Fig. 17. Status of bond 1 and bond 2 in a H-cell along a biaxial loading path, in terms of the H-cell
orientation (45 deg, 47 deg, 49 deg, 53 deg, 55 deg, 71 deg).

Figure 17 depicts the bond failure chronology for a H-cell oriented in a certain direction along the loading path. The response of the H-cell in the direction (45 deg) is marked by the failure of bonds in compression as shown in the domain C (Fig. 16). However, the H-cells in the directions 47 deg and 49 deg, 53 deg experience the failure of bond 1 in compression first and then the failure of bond 2 in traction, which is consistent with the existence of domain B. The response of H-cells along the directions 55 deg and 71 deg is dominated by the failure of bond 2 in a traction regime, as observed

#### 440 in domain A.

The evolution of the meso-strains for the H-cells in the direction (53.5 deg) is shown 441 in the Fig. 18. It can be seen that the mechanical behavior of the H-cells is in 442 compression first, whatever the meso-strain ( $\varepsilon_n$ ,  $\varepsilon_t$ ). The meso-strain is derived from 443 the macroscopic strain, and a sharp volume change is observed at the axial strain of 444 0.86%, due to the massive bond failure in traction. Thus, the meso-strain also 445 experiences a sharp evolution, especially for the meso-strain  $\varepsilon_n$  with a transition from 446 a compression regime to an extension regime, while the meso-strain  $\varepsilon_t$  is still in a 447 compression regime. Fig. 19 shows the response of contact forces along a biaxial 448 loading path. The normal contact forces  $(F_{n1}, F_{n2})$  are in compression first, as observed 449 for the meso-strains. At the axial strain of 0.86%, the meso-strain  $\varepsilon_n$  vanishes and 450 takes negative values, indicating that an extension regime occurs. As a result, the 451 normal contact force  $F_{n2}$  becomes a traction force. However, the normal contact force 452  $F_{n1}$  remains in compression, as the meso-strain  $\varepsilon_t$  still corresponds to a compression. 453 In Fig. 19, the normal contact force  $F_{n1}$  reaches the compression strength limit, 454 455 causing the failure of bond 1 in compression. The residual contact law is activated and the tangential contact force  $F_{t1}$  reaches the Mohr-Coulomb friction limit. Then, the H-456 cell meso-stress vanishes once the bond 2 brakes in traction. 457 458



460 Fig. 18. Evolution of meso-strain components  $\varepsilon_n$  (relative length variation in the H-cell orientation) 461 and  $\varepsilon_1$  (relative length variation perpendicular to the H-cell orientation) for the H-cell in the direction 462  $\theta = 53.5 \text{ deg}$  along a biaxial loading path.



465 **Fig. 19.** Evolution of contact forces within the H-cell oriented in the direction  $\theta = 53.5$  deg along a 466 biaxial loading path (a); evolution of the contact force for bond 1 within the Mohr-Coulomb plane (b).

Figure 20 shows the mechanical behavior of the H-cells in the direction  $\theta = 49$  deg. The H-cell is in contraction first along both axial and lateral directions. Then the mesostrain  $\varepsilon_n$  gradually decreases and reaches an extension regime. Fig. 21b shows that the normal contact force  $F_{n1}$  reaches first the compression strength limit, making bond 1 fail. Meanwhile, the normal contact force in bond 2 evolves from a compression regime to a traction regime until the bond fails.





476 Fig. 20. Evolution of meso-strain components  $\mathcal{E}_n$  (relative length variation in the H-cell orientation) 477 and  $\mathcal{E}_1$  (relative length variation perpendicular to the H-cell orientation) for the H-cells in the direction 478  $\theta = 49 \text{ deg}$  along a biaxial loading path.





481 Fig. 21. Evolution of the contact force for bond 2 in the H-cell oriented in the direction  $\theta = 49 \text{ deg}$ 482 along a biaxial loading path (a); evolution of the contact force for bond 1 within the Mohr-Coulomb 483 plane (b).

#### 3.5 Limitations of the *H*-bond model 484

The H-bond model is an example of micromechanically-based model, departing from 485 usual phenomenological models that require sophisticated formulations to capture the 486 complexity observed on the macroscopic scale. Micromechanically-based models 487 assume that the complexity stems mainly from the structural intricacy (geometrical 488 complexity), whereas the local contact law between particles can be formulated in a 489 very straightforward manner. In this perspective, the local constitutive relations taking 490 place at contacts in the H-bond model are very simple, without any phenomenological 491 sophistication, if at all possible. This simplicity can of course be debated in the specific 492 case of the H-model, as the microstructural diversity cannot be approached as precisely 493 494 as it can be done when using a Discrete Element Method. Indeed, the H-model assumes

that the granular assembly is simply described by a spatial distribution of independent 495 hexagonal grain patterns. Each hexagon, oriented along a given direction of the physical 496 space, contributes to the total stress existing within the assembly. However, the different 497 hexagons do not interact with each other. It should also be noted that, compared with 498 cohesionless granular materials, typical mesostructures found in (2D) cohesive granular 499 materials can be larger than hexagons (cohesive bonds between particles contribute to 500 a better mechanical stability of mesostructures containing larger numbers of grains), 501 502 which results in longer correlation lengths within the assembly. It is clear that more complex criteria can be selected to describe bond failure. In this manuscript, it was 503 assumed that the three failure modes (traction, compression and shear) are independent, 504 disregarding more complex, coupled formulation including shear 505 and traction/compression effects within a unique plastic limit surface. In addition, we have 506 adopted the same strengths for all the bonds, which may result in more abrupt failure 507 patterns (as observed in Figs. 13 and 14). This first approach was necessary to have a 508 first overview of the model capability, before favoring a more complex, random 509 510 distribution of strengths for the different bonds.

511 Furthermore, as the grains themselves constitute the microscopic (i.e. smallest) scale, 512 we disregard any processes taking place at a smaller scale. Thus, the likely irregular 513 shape of grains after bond failure cannot be captured by the model. This simplification 514 is mainly debatable when dealing with cemented granular materials, where skeleton 515 grains are immerged within a cemented matrix. If we restrict the current approach to 516 granular assemblies with independent solid bonds between grains, this approximation 517 is probably much more reasonable.

In spite of these inherent limitations, the approach stands as a first attempt to address the case of bonded granular assemblies, by developing a full micromechanically-based model. Even though the above-mentioned limitations should be carefully considered in future works, the results presented in this manuscript suggest that this approach can be deemed as a powerful alternative to standard phenomenological approaches.

## 523 4. Conclusion

In this paper, the behavior of cemented granular materials is investigated by considering the extended *H*-bond model and the underpinning microstructural mechanisms taking place within bonds. The model represents the granular assembly by a distribution of hexagonal patterns of contacting grains with solid bonds.

Each bond is described by a triad including two spherical particles linked by an intermediate beam-like bond, in which three failure modes (compression, traction, and shear) can occur. Once a bond has failed, it is supposed not to exist any longer, and the corresponding triad is replaced with a standard, unbonded pair of grains in elastofrictional interaction.

533 By simulating biaxial loading paths and proportional strain loading paths, the 534 interplay between the different bond failure modes has been explored. It is shown that 535 the bond has an effect mostly in the early stage of loading, where a higher deviatoric 536 stress peak occurs with a more pronounced, subsequent softening regime. As expected, the increasing bond strength contributes to an increase in this peak stress. Furthermore, depending on the bond strength in compression, tension and shear, the bonds are likely to break along preferential orientations. Progressively, the stress-strain response curve approaches the one that would have been obtained for a cohesionless specimen. These results are consistent with the conclusion drawn by Jiang et al. (2013).

542 When proportional strain loading paths are considered, it is shown that the bonds are 543 also persistent at the beginning of the loading, whatever the volumetric strain regime 544 (contractant, dilatant or isochoric). Likewise, depending on the bond strength in 545 compression, tension and shear, the bonds fail in compression regime first, as the more 546 loaded bonds are in a compressive regime at the beginning of the loading. Then, the 547 contacts open when the tensile strength is reached.

548 More interestingly is the relation observed by tracking the process of bond failure 549 during the loading paths. The evolution of both the bond failure ratio and the bond 550 failure rate along the biaxial loading reveals that the peak of the bond failure rate 551 coincides with the peak stress, and is followed by a sharp, subsequent decrease. As the 552 bonds break in traction, the corresponding H-cells vanish, resulting in fewer cells 553 involved in the macroscopic stress. Hence, a remarkable reduction in the macroscopic 554 stress occurs, leading to a noticeable softening.

Analyzing the microscopic mechanical behavior offers more insight to understand the macroscopic response. Along biaxial loading paths, both bond 1 and bond 2 in a Hcell experience failure in a compression regime along the axial loading direction, while the bonds are gradually dominated by a traction or a shear regime, leading bonds to a tensile or shear failure in the lateral direction. For a particular range of directions, bond 1 first fails in a compression regime leading to the change in contact stiffness, which results in an earlier failure of bond 2 in traction.

562 Finally, the *H*-bond model is versatile and its micromechanical description makes it 563 possible to investigate at the intermediate scale the origins of various macroscopic 564 properties.

# 565 **CRediT authorship contribution**

Zeyong Liu: Investigation, Simulation, Validation, Methodology, Writing – original
draft. Francois Nicot: Supervision, Conceptualization, Writing – review & editing.
Antoine Wautier: Supervision, Methodology, Writing – review & editing. Felix Darve:
Writing – review & editing.

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579

# 580 Appendix A

In the compression regime, once bond is broken, the triad reduces in a single contact 581 between two particles. The normal contact force is described by an elastic relation, with 582 a stiffness different from that prior failure (Fig. 22). Indeed, for a serial bond contact 583 model, a triad (particle-bond-particle) is composed of three springs. Before bond failure, 584 the normal contact force evolution is given by equation (1), with a stiffness reported in 585 equation (3). Upon bond failure, the normal contact force evolution is given by equation 586 (5), and the stiffness is reported in equation (7). It can be seen (Fig. 22) that after bond 587 failure, a total unloading until zero normal force leads to a residual normal displacement 588 (u - u'), where u' stands as the reversible part of the total normal displacement u. 589 590 The displacement u of the triad before bond failure is given by:

591 
$$u = \frac{F_n}{k_n}$$
(A1)

592 whereas the reversible part u' of the displacement expresses as:

593 
$$u' = \frac{F_n}{k'_n}$$
(A2)

The residual part (u - u') is the consequence of the triad damage, associated with a change in the total stiffness (Grassl and Jirásek, 2006). By combining equations (A1) and (A2), it can be obtained:

597 
$$u - u' = F_n \left( \frac{1}{k_n} - \frac{1}{k'_n} \right)$$
 (A3)

598 Combining equations (3) and (7) expressed with the relationship involving the total 599 stiffness and the stiffness of particles  $(k_{np})$  and bond  $(k_{nb})$  in a triad, yields the 600 following equation (A4):

$$601 u - u' = \frac{F_n}{k_{nb}} (A4)$$

where the residual normal displacement (u - u') is expressed as a function of the normal stiffness of bond  $k_{nh}$ .

It should be noted that  $u_b = u - u'$  corresponds to the elastic part of the normal displacement within the bond. After failure, this part of the displacement will not be recovered.



607

608 Fig. 22. Evolution of normal contact forces for a bonded contact, prior and after bond failure.

609

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